

Surface Modification of Detonation Nanodiamonds with Platinum and Palladium Nanoparticles

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Abstract

Acetylacetonates of platinum group metals ($M(\text{acac})_n$, $\text{acac} = \text{CH}_3\text{COCHCOCH}_3$, n - oxidation state of metal) are the most suitable precursors for use in highly effective MOCVD (metal-organic chemical vapor deposition) processes because of their thermal properties, low price of initial ligand and accessible synthesis methods. In this study we have developed a simple, scalable inexpensive, MOCVD-like approach to the deposition of Pt and Pd nanoparticle dispersions on powder detonation nanodiamonds (DND) using combustion in air of powder mixtures of DND with Pt and Pd acetylacetonates. This one-step process requires no specialized apparatus, and is conducted at mild temperatures ($180^\circ\text{C} - 250^\circ\text{C}$). The substrate surface of DND initiates chemical thermal destruction of precursor and deposition of nanoparticle networks composed of constituent particles. DND-supported Pd and Pt nanoparticles with their loading being from 10 and 5wt. % were obtained the average of particle size 20-25 nm and 7-10nm decreases with decreasing the metal loading. For additional purification and disaggregation of the commercial detonation-synthesized nanodiamond (DND_{comm}) the method of annealing of the powder mixtures DND_{comm} with $\text{Na}(\text{acac})$ have been used. The purified diamond nanoparticulates (DND), as well as DND_{comm} , Pt/DND and Pd/DND, were subjected to physicochemical characterizations, such as X-ray diffraction (XRD), scanning electron microscopy (SEM), and energy dispersive X-ray (EDX) analysis.

Keywords: detonation-synthesized nanodiamond supports, platinum and palladium nanoparticles, combustion, chemical modification, DND additional purification

1. Introduction

Detonation-synthesized nanodiamond is a low-dimensional carbon nanomaterial, along with fullerene nanotubes and nanofibers. It possesses the properties of a diamond, for example, the outstanding hardness and antiwear, as well as the features of a nanoscale material, such as an ultrafine particle size and a giant specific surface area as well as a large number of surface defects (Dolmatov, 2007; Mochalin, Shenderova, Ho Dean & Gogotsi, 2012). DND with a large specific surface area (Spitsyn et al., 2010) has great potential in the development of new types of composite materials (Gibson et al., 2009), especially as a substrate for hosting metal nanoparticles (Panich et al., 2009). Recent works showed a number of potential DND applications. One of the most important is use nanodiamonds as a platinum group metal catalyst carrier, which is intended for application in fuel cell electrodes, electrochemical gas sensors and other electrochemical devices. The efficiency of the detonation nanodiamonds as a support for catalytic metals of platinum groups has been stated (Kachevskii, Golubina, Lokteva & Lunin, 2007; Vershinin et al., 2011; Magdalinova, Kalmykov & Klyuev, 2014; Fang Gao, Nianjun Yang & Nebel, 2013).

The conventional techniques for preparing nanodiamond as a catalyst carrier of Pt-metals are wet impregnation and chemical reduction of salts/acids metal precursors (H_2PtCl_6 , H_2PdCl_4 , RhCl_3) (Vershinin & Efimov, 2009; La-Torre-Riveros et al., 2011), alternative synthesis methods is based on microwave-heating of H_2PtCl_6 solution (L.Y. Bian, Y.H.Wang, J.B. Zang, F.W. Meng & Y.L. Zhao, 2012). One of the most important drawbacks in the classical preparation of wet procedures is the stepwise nature of the process, in which a slight change in any of the parameters, at any stage, may lead to drastic changes in the characteristics and performances of the resulting

materials. In contrast to conventional multi-step wet chemistry techniques, the MOCVD method (metal-organic chemical vapor deposition) (Serp et al., 1999; Vahlas, Juares, Feurer, Serp & Caussat, 2002; Aksoylu et al., 2003; Vahlas, Caussat, Serp & Angelopoulos, 2006) and flame technology (Strobel, 2009; Choi, Lee H, Shim & Lee D., 2010; Sudong Yang et al., 2014) allow preparation of supported metal nanoparticles of platinum group metals onto nanocarbon supports in a single step. MOCVD methods and traditional flame synthesis rely on volatile metal precursors. Now the great number of volatile metalorganic Pt-metal complexes are known, the analysis of their thermal properties is presented in (Vargas Garcia & Takashi Goto, 2003; Morozova et al., 2001). In a number of recent works in obtaining supported on nanocarbon carriers Pt and Pd nanoparticles volatile acetylacetonates of these metals were used. In (Papandrew et al., 2013; Papandrew et al., 2013; Alexander B. Papandrew, Calum R.I. Chisholm, Ramez A. Elgammal, Mustafa M. Özer & Strahinja K. Zecevic, 2013) platinum nanoparticles with sizes approximately 3 nm were deposited on carbon nanotubes by metal-organic chemical vapor deposition of platinum acetylacetonate. For the production of supported on carbon agglomerates Pt particles (2-7nm) Pt(acac)₂-containing xylene droplets are injected into the acetylene-air flame (Choi, Lee H, Shim & Lee D., 2010). The graphene-supported monodisperse Pd nanoparticles were prepared by simultaneous reduction of graphene oxide and Pd(acac)₂ in flame technology so-called flame-spray pyrolysis (Sudong Yang et al., 2014).

In our studies (Isakov, Lyamkin & Isakova, 2006; Isakova, Petrakovskaya, Isakov, Bayukov & Velikanov, 2006) it has been shown for the first time that the powder mixture of detonation nanodiamonds with metal acetylacetonates ignites spontaneously when heated in air at temperatures lower than 200°C. A similar reaction was observed by us for the powder mixtures of metal acetylacetonates with fullerite and taunitom (Isakova et al., 2013). Platinum metal acetylacetonates separate out metal in the condensable phase. Moreover, the addition of metal acetylacetonate in the diamond-carbon detonation charge (DC-charge) initiates spontaneous combustion and low-temperature combustion in mode of smoldering, as a result bulk of "non-diamond" fraction in the DC-charge is burned. After subsequent liquid-phase acid purification of residuum selected DND has a higher purity than DND, purified by conventional methods of liquid-phase purification (Isakova & Isakov, 2004).

In this study, Pt and Pd nanoparticles were deposited on DND powders using self-developed one-step synthesis method i.e. the combustion in air of solid-phase mixtures of DND with Pt and Pd acetylacetonates. To remove a non-vanishing amount of residual metal impurities in commercial detonation nanodiamonds (DND_{comm}) the mixture of DND_{comm} with sodium acetylacetonate was annealed.

2. Experimental Part

DNA_{comm} produced by the Altai Federal Scientific and Production Center (Biisk), was used as an object of study. Metal acetylacetonates were obtained using the known techniques. Pd(acac)₂ was prepared by the method which is described in (Filatov, Basova & Igumenov, 2008). Platinum acetylacetonate Pt(acac)₂ was synthesized in the Institute of Inorganic Chemistry of SB RAS (Novosibirsk) according to the method (Igumenov, Mikheev, Zherikova, Morozova & Zharkova). Salts were purified from impurities by the sublimation method.

The powder mixtures of the purified detonation nanodiamond (DND) with Pt (acac)₂ or Pd (acac)₂ in an amount of 5 and 10 wt.% of the metal relative to DND were placed in flat-bottomed crucibles and heated on electric stove at temperature ~ 250°C up to spontaneous ignition of powder mixture, which is observed at temperature ~ 180°C, and then heating was stopped, the process proceeded spontaneously in the smoldering regime with increasing temperature up to ~ 250°C. The smoldering temperature of powder mixtures was measured by a thermocouple (accuracy ±5 °C). For the obtaining of the purified detonation nanodiamond the powder of DND_{comm} was ground in a mortar with Na(acac) (~20 wt. %), and the obtained mixture was heated up to the smoldering combustion. With finishing of the combustion, the obtained powder was boiled in HCl solution and the color solution was removed after settling. The residuum was washed with distilled water and dried in air at 90 °C.

The characterization of the prepared samples of DND, Pt/DND and Pd/DND was performed by X-ray diffraction (XRD) (Bruker D8 Advance), energy dispersive X-ray (EDX) analysis (LEO-420 and INCA x – sight), scanning electron microscopy (SEM) (Hitachi S5500), and particle size distribution measurements (CPS Disc Centrifuge Model DC 24000).

3. Results and Discussion

According to (Tsiganova & Dyagileva, 1996). Reactivity of β- diketonates in the reaction of thermal decomposition in the case of metal acetylacetonates thermal decomposition in a vacuum into the gas phase either thermally stable Hacac or acac- radical is released. Oxygen reduces the temperature of M(acac)_n disintegration and can affect the composition of the condensed phase, in the gas phase Hacac and acac oxidation products predominate CO and CO₂.

When heating the powder mixtures of DND with $\text{Pt}(\text{acac})_2$, $\text{Pd}(\text{acac})_2$ and $\text{Na}(\text{acac})$ in air, the spontaneous ignition of the mixtures occurs at 180- 200°C, and then the combustion process develops over the entire reaction mixture in the smoldering regime. In the absence of DND, the platinum and palladium acetylacetonates, being volatile (Vargas Garcia & Takashi Goto, 2003; Morozova et al., 2001), are completely sublimated within the temperature range where the spontaneous ignition of their mixtures with DND occurs, while $\text{Na}(\text{acac})$ does not undergo substantial changes. The average ignition temperature of the DND mixtures with $\text{M}(\text{acac})_n$ is lower than the temperature of their own pyrolysis of the metal salts (Tsiganova & Dyagileva, 1996), indicating the catalytic character of the reaction. In this case, the mechanism of DND catalytic action is most likely because of their ability to adsorb oxygen on the surface.

The exothermal chemical reaction in the $\text{Pt}(\text{acac})_2/\text{DND}$ and $\text{Pd}(\text{acac})_2/\text{DND}$ systems proceeds on the interface of solid DND particles with metal salt vapor. The mass transfer is made by the metal salt vapors. The role of the metal acetylacetonate in the spontaneous combustion, we believe, comes down to that on the bottom surface, due to the adsorbed oxygen, they accelerate the thermal decomposition reaction of $\text{M}(\text{acac})_n$ and the exothermic oxidation of released Hacac and acac. In addition, it is possible, that the process becomes autocatalytic nature by the catalytic action of precipitated metal particles.

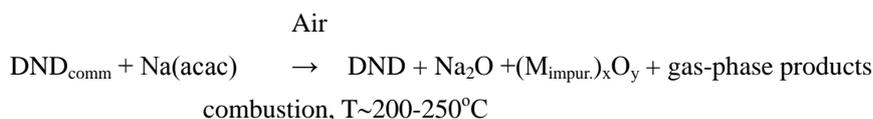
The reaction between the $\text{Na}(\text{acac})$ powders being mixed with DND proceeds on the solid particle interface. According to the evidence observed, the process of DND combustion with the metal acetylacetonates can be attributed to the OSHS (organic self-propagating high temperature synthesis) processes described in literature (Klimchuk, 2000).

A relatively low temperature of the thermal process, regardless of the red glow being observed, can be accounted for by small-sized local regions of overheating (hot spots) appearing upon catalytic thermo-oxidative destruction of $\text{M}(\text{acac})_n$, where the exothermal reaction occurs. Detonation nanodiamonds generally have low thermal conductivity (Vereschagin, 2005) due to the gases being adsorbed on the particle surface. Therefore, while the heat is released in the hot spots, DND does not have enough time to heat up and oxidize. Thus, the average temperature of the reaction mixture can be lower than that in the hot spots.

Detonation-synthesized nanodiamond particles are enclosed in amorphous carbon or graphite. Even if the particles are purified, the metal impurities can be found inside the nanodiamond aggregates, so the nanodiamond aggregates should be disintegrated to remove the trapped impurities (Kulakova, 2004).

To additionally purify DND_{comm} from metal impurities the mixture of DND_{comm} with $\text{Na}(\text{acac})$ was annealed in air in the combustion regime (Scheme 1) and after annealing the desorbed metal impurities were transformed into soluble salts.

Scheme 1: Additionally purify DND_{comm} from metal impurities.



Energy dispersive X-ray spectra (EDX) of the DND_{comm} and DND samples are summarized in Fig. 1a and Fig. 1b, respectively, and data on their chemical composition are given in Table 1. As can be seen, the DND_{comm} sample contains besides C some impurities of O, Al, Si, Fe. Only C and O are detected for the DND sample. The oxygen content is reduced (see Table 1). This may be related with changes in the composition of the DND surface groups.

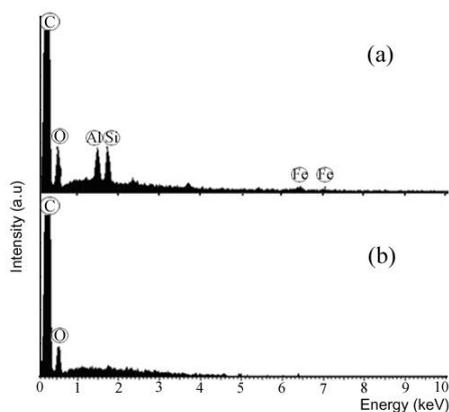


Figure 1. The energy dispersive X-ray spectrum: a) DND_{comm}, b) DNDTable 1. The elemental analysis of DND_{comm}, DNA

Element	Wt. %		Atom. %	
	DND _{comm}	DNA	DND _{comm}	DNA
C	87.55	91.32	90.97	93.34
O	10.58	8.68	8.25	6.66
Al	0.74		0.34	
Si	0.8		0.36	
Fe	0.33		0.07	

The EDX spectrum (Fig.2) of the combustion product of DND/ Pt(acac)₂ (5 wt.% Pt relative to the nanodiamond) shows only C and Pt. Oxygen is not detected, likely due to its low content. The platinum content (see Table 2) in the Pt/DND sample (4.94%) corresponds to the nanodiamond/platinum ratio in the initially prepared mixture of Pt(acac)₂/DND. The measured masses of the combustion products of the DND/Pt(acac)₂ and DND/Pd(acac)₂ mixtures correspond to the total weight of DND and the metal in the initial mixtures.

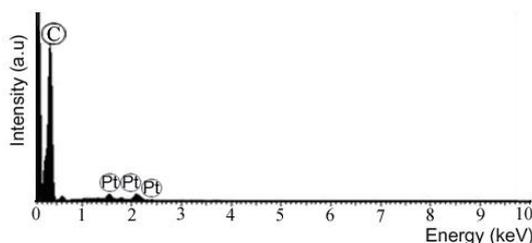


Figure 2. The energy dispersive X-ray spectrum, Pt/DND

Table 2. The elemental analysis of Pt/DND

Element	Wt. %	Atom. %
C	95.06	99.68
Pt	4.94	0.32

The powder XRD patterns for the Pt/DND and Pd/DND samples with the metal loading of 5 wt.% and 10 wt.%, are shown in Fig. 3, 4.

The XRD patterns confirm the presence of crystalline Pt and Pd nanoparticles with characteristic peaks at 2θ values of (Pt): 39.8°, 46.2°, 67.4°, 81.2°, 85.8° (Fig. 3); and of (Pd): 40.1°, 48.7°, 68.2°, 82.2°, 86.6° (Fig. 4), corresponding to the reflection planes (1 1 1), (2 0 0), (2 2 0), (3 1 1), (2 2 2) respectively. It could be attributed to face-centered cubic Pt and Pd. The peaks appear at 2θ 33.9°, 54.7°, 71.4° (Fig. 4) corresponding to PdO. The partial formation PdO (~2-3%) is likely to be due to Pd high reactivity as compared to Pt. The peak broadening evidences the nanoscale dimensionality of the Pt, Pd and PdO particles.

The diffraction peaks observed at 2θ 43.98°, 75.28°, and 91.68° are attributed to the cubic diamond planes (1 1 1), (2 2 0), and (3 1 1), respectively. With the Pt and Pd loading increasing, the intensity of the diffraction peaks of the nanodiamond decreases. This implies more metal nanoparticles being deposited on the nanodiamond surface.

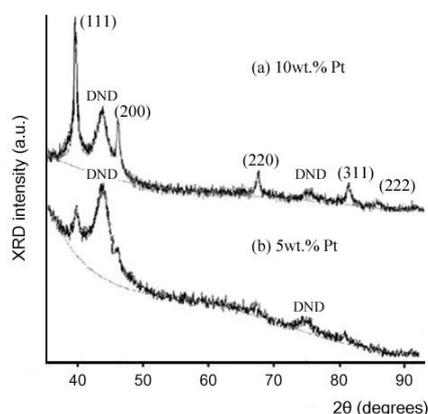


Figure 3. The XRD patterns of Pt/DND: a) at 10wt.% Pt; b) at 5wt.% Pt

The crystallite size of Pt and Pd is found to be dependent on the metal loading, indicating better dispersion of the Pt-metal particles on the DND support at lower Pt-metal loading.

For the metal loading of 10wt.% and 5 wt.%, the average Pt and Pd crystalline particle sizes evaluated using the Scherrer method were about 25nm and 10nm, respectively.

The simultaneous loading of Pd and PdO nanoparticles on the DND surface is of interest. For example, it is shown (Aramendia et al., 1999; Golubina, Lokteva, Majouga, Lobanov & Lunin, 2011; Magdalinova, Kalmykov & Klyuev, 2012) that upon the formation of a high-effective Pd catalyst the structure of the active center is to contain both the reduced and the oxidized metal. However, in the common Pd catalyst preparation processes the reduction stage is always present, with a partially oxidized metal being difficult to obtain in this case.

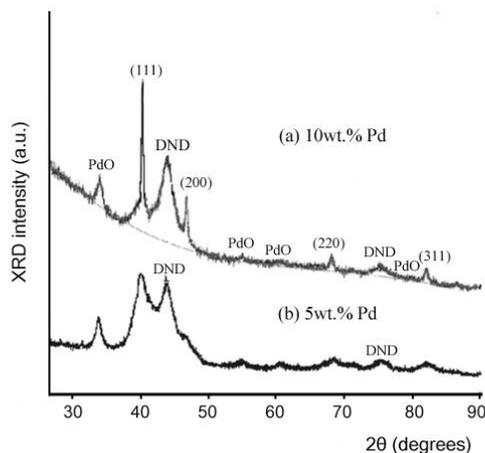


Figure 4. The XRD patterns of Pd/DND: a) at 10wt.% Pd (PdO- 2,07%); b) at 5wt.% Pd

It is known that nanodiamonds prepared by explosive detonation have primarily the particles, 4~6 nm in diameter, but they undergo severe aggregation, forming aggregates of hundreds of nm, even several μm in diameter, which can withstand ultrasonic treatment (Kriiger et al., 2005). The details of the surface morphology of the DND_{comm} , and the prepared DND, Pt/DND, Pd/DND samples were obtained using scanning electron microscopy (SEM).

SEM images (at a magnification of 30000) of DND_{comm} . DND and DND after sonication are presented in Fig. 5. It can be seen from the micrographs that the DND_{comm} . aggregates with diameter more than 1 μm (Fig. 5a) can form DND aggregates with diameter less than 200-250 nm after annealing with Na(acac) (Fig. 5b) and the DND aggregates are broken down to aggregates of about 30-10nm after the sonication (Fig. 5c)

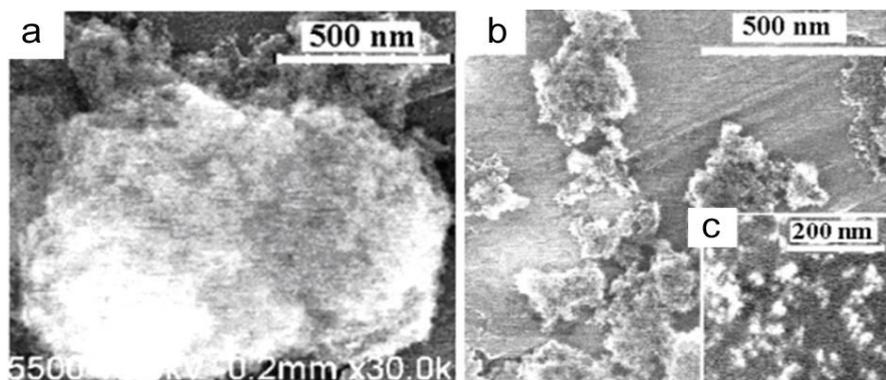


Figure 5. The micrographs: a) the aggregated DND_{comm} particles, b) the aggregated DND particles, c) the aggregated DND particles after sonication

The particle-size distribution curves of DND_{comm} after sonication (Fig 6a in the inset) as well as the DND particles after sonication (Fig. 6b) show a strong difference. The size of the DND particles is significantly reduced in comparison with the size of the DND_{comm} ones. Their mean particle diameters are 8-10 nm and 200-500 nm, respectively. The particle-size distribution curves of Pd/DND (5 Wt.% Pd) are shown in Fig. 6c. After loading Pd on the DND surface the size of most particles of the obtained composite is about 5-7 nm. It is less, than DND_s particles size. It's possible, that aggregated DND particles destruction occurs more intensive in the

smoldering process of DND+Pd(acac)₂ powder mixture, then under ultrasonic treatment.

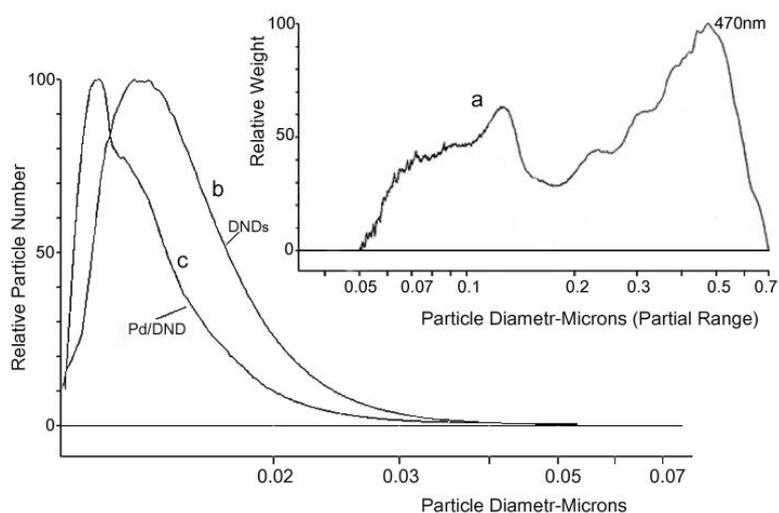


Figure 6. The dependence of the relative distribution of the particle amount on the sample size: DND_{comm} – curve a; DND - curve b, Pd/DND- curve c

Figure 7 presents the micrographs of DND, Pt/DND and Pd/DND at a magnification of 300000. The formation of the DND particle aggregates of about 200 nm was registered (Fig. 7a). The structure of the Pd/DND and Pt/DND surfaces reveals different morphology. As is seen, the Pd/DND and Pt/DND particles appear to have grain-like morphology, with the grains being 20-5 nm in size, depending on the metal loading on the DND support. With 10 wt.% of metal in the Pt /DND (Fig. 7b) and Pd /DND samples (Fig. 7c) the grain size is equal to ~ 20 nm and smaller. In the Pd /DND sample with 5 wt. % Pd (Fig.7d) the grain size is ~7nm and smaller (see also Fig. 6c).

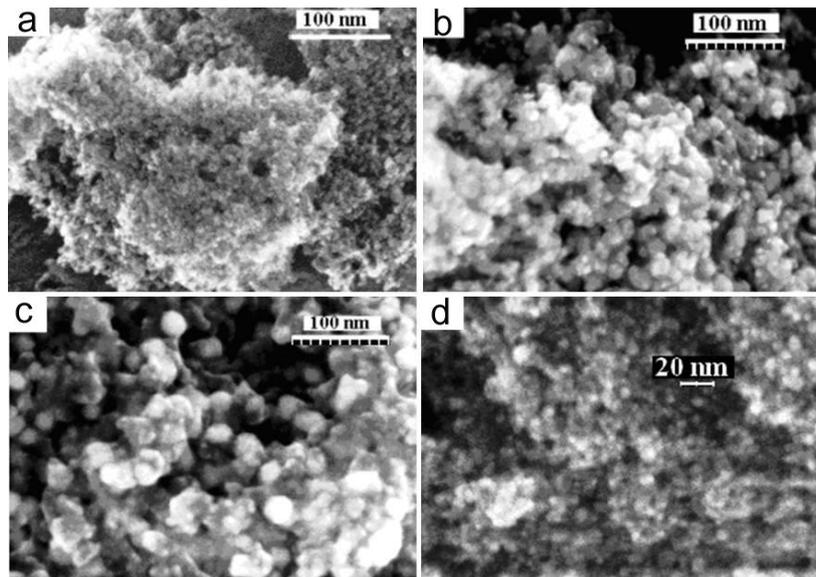


Figure 7. The micrographs: a) the aggregated DND particles, b), c) DND /Pt, DND /Pd (10wt% of the metals), d) DND /Pd (5wt% Pd)

4. Conclusion

A one-step synthesis method to prepare Pt and Pd nanoparticles supported on the surface of detonation nanodiamond (DND) powders has been used. Metal nanoparticles were deposited on the surface of DND using the combustion in air of the powder mixtures of DND with volatile acetylacetonates of Pt and Pd. The density growth of the Pt and Pd nanoparticles on the DND support could be controlled by manipulating the mass ratios between the Pt and Pd precursor and the nanodiamond support. Scanning electron microscopy, and X-ray diffraction patterns confirm the formation of the Pt and Pd nanoparticles and the increase of the average particle

sizes ($\approx 5\text{--}25$ nm) with increasing the metal loading. The partial oxidation of Pd ($\sim 2\text{--}3\%$) is likely to be due to its high reactivity as compared to Pt. The thermal destruction of Pt and Pd acetylacetonates offers a simple route to prepare the dispersed Pt and Pd catalyst nanoparticles on DND surface for their possible application as an advanced electrode material in direct methanol fuel cells. Further research is needed to reduce the particle size. Annealing the commercial detonation-synthesized nanodiamond with the addition of Na acetylacetonates is shown to allow desorbing metal impurities which are difficult to remove during the purification of detonation nanodiamonds using conventional techniques. The annealing process is also found to allow breaking down the DND aggregates.

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